# Structure of niobium to 365 GPa using ultrafast x-ray diffraction and shock compression

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The phase stability, crystal structure, and melting of Nb have been examined under high pressure shock compression to 365 GPa using ultrafast *in situ* x-ray diffraction measurements on two x-ray free electron laser facilities. On compression, Nb remains stable in the bcc phase up to 220 GPa, with coexistence of bcc-Nb and liquid from 249 to 298 GPa, and complete melting at 301 GPa, with melt identified by diffuse liquid diffraction. Melting initiates at higher pressure than expected based on theoretical predictions, and the data consistently excludes the presence of other suggested phases of Nb at these pressures, including *Pnma* and hcp, thereby resolving the long-standing structural discrepancy. Singh-type strength analysis provides experimental evidence for a change in sign of the elastic anisotropy parameter at 121 GPa, consistent with the previously reported heat-induced hardening to heat-induced softening transition in Nb.

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## I. INTRODUCTION

Niobium (Nb) is a group VB metal situated between zirconium and molybdenum in the periodic table. Because of its high melting point, chemical inertness, and strength properties, Nb is widely used in industrial applications. These applications include high-entropy alloys [1,2], superconducting magnets for fusion research [3], medical imaging [3], and energy storage technologies [4,5]. Nb exhibits unusual elastic behavior, including atypical softening [6], under high pressure [7–9], which has motivated extensive studies at room temperature using both experimental [6,10–15] and computational techniques [9,16–19]. Given niobium's role in high-performance alloys and fusion reactors, surprisingly few studies have explored its behavior under high pressure and temperature [20–23], in contrast to other body centered-cubic (bcc) metals such as V [24–28], Mo [24,29–33], Ta [34–39], and W [24–26,40,41].

Early shock compression work on Nb, by Al'tshuler *et al.* [42], reported a second-order phase transition at ∼94 GPa, as inferred from a change in the gradient of the shock-velocity versus particle-velocity curve (Supplemental Material Sec. A [43]). Further studies failed to reach sufficient pressures to observe this phase transition [25,26,67], while subsequent molecular dynamics simulations along the principal Hugoniot predicted a solid-solid phase transition from bcc to hexagonal close-packed (hcp) at ∼75 GPa [68]. This pressure is consistent with the dramatic stiffening of the $C_{44}$ elastic constant in Nb previously reported by Koči *et al.* [7] at 300 K and has been considered as the driving mechanism for structural changes in V at high pressure [69]. More recently, Errandonea *et al.* [23], using static-compression techniques, combined laser-heating x-ray diffraction (XRD) and computation to investigate the phase stability, including melting, to ∼60 GPa and ∼3300 K on Nb. They reported a solid-solid transition from bcc to an orthorhombic *Pnma* structure some 1000–1500 K prior to melting. Extension of the experimental results to higher pressures using *ab initio* molecular dynamical simulations suggested the transition to the orthorhombic phase would occur at ∼142 GPa along the principal Hugoniot [23]. The existence of intermediate solid phases of Nb and the onset of melting remain unresolved, with conflicting predictions and static results, motivating our ultrafast shock compression measurements.

Significant progress in generating and studying extreme pressure-temperature (*P*-*T*) states through dynamic compression and XRD has been made over the past decade [36,70–75]. In some instances, these methods couple long pulse (∼10 ns), high-energy drive lasers with the extreme brightness of modern x-ray free-electron lasers (XFELs) [76–79]. The new DiPOLE 100-X laser [80,81], at the High Energy Density (HED) instrument of the European x-ray Free Electron Laser (EuXFEL) [82], and the long-pulse laser system at the Matter in Extreme Conditions (MEC) end station of the Linac Coherent Light Source (LCLS) [83], enable ultrafast *in situ* XRD studies along the principal Hugoniot with fine temporal control.

In this paper, we apply these technologies at HED and MEC to shock compress Nb to extreme states and carry out *in situ* crystal structure investigations using XRD with femtosecond time resolution, resolving the structural controversy of Nb along the principal Hugoniot. We present diffraction data for solid-bcc and liquid Nb up to a maximum pressure of 365(32) GPa. The data set comprises twenty two shots from solid Nb, six shots along the melt curve showing coexistence of solid and liquid phases, and two shots from fully molten Nb. We observe clear evidence of the incipient melting of Nb at a pressure of 249(41) GPa, as determined from a reduction in Bragg peak intensities and the appearance of a strong diffuse scattering signal from the liquid phase. Prior to the onset of melting, we find no evidence of the previously reported bcc → hcp or bcc → *Pnma* solid-solid transitions [23,68]. Our results establish the structural response of Nb along the principal Hugoniot, quantify the melt boundary, and reveal a change in sign of the elastic anisotropy parameter, *S*.

## II. EXPERIMENTAL METHODS

The experiments were carried out during two campaigns at different XFEL facilities. The first was conducted in

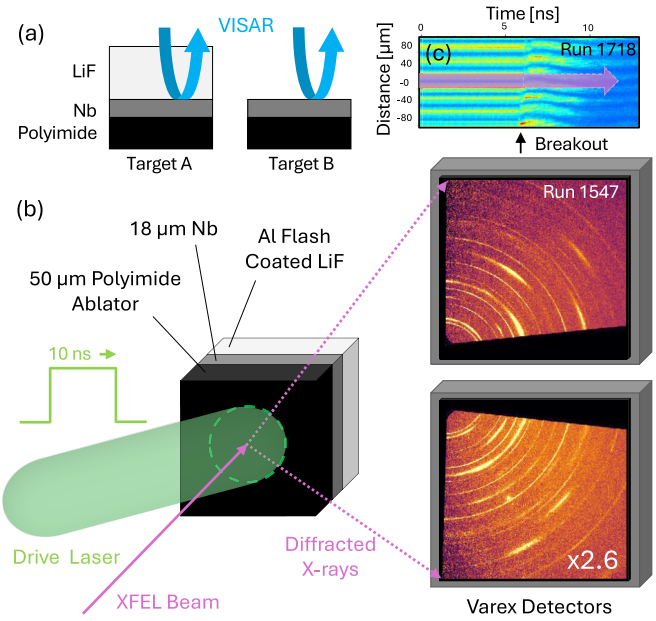



FIG. 1. (a) The two target types shot during this experiment indicated by targets A and B. (b) Schematic of the experimental setup used to collect VISAR and *in situ* XRD data at HED and MEC, not to scale. VISAR measured the particle velocity at the Al-LiF interface of the A-type targets. The two Varex detectors on HED enable approximately half of the Debye-Scherrer XRD pattern to be collected, and an example 2D diffraction image from laser-shocked Nb at 82(26) GPa (Run 1547 HED) is shown with ×2.6 magnification. (c) Representative VISAR image for a type-A target (Run 1718 HED). The pink arrow indicates the width of the XFEL beam in the vertical direction at HED (showing the shock was planar over the probed region) and $t = 0$ ns represents the time at which the drive laser was turned on.

Interaction Chamber 2 (IC2) at the HED instrument [82] of the EuXFEL during community campaign #6659. The second was conducted at the MEC end station [83] of the LCLS, during campaign L-1019923. A schematic of the standard experimental setup on HED is shown in Fig. 1. Pressure determination using the velocity interferometer system for any reflector (VISAR) [84–86] is described in Supplemental Material Sec. C [43].

### A. Target design

We employed two sets of targets during the HED campaign: a windowed configuration of polyimide-Nb-lithium fluoride (LiF) (target A), and a free-surface polyimide-Nb (target B), as shown in Fig. 1(a). On the laser-drive side, for both targets types, a 50(2) μm-thick polyimide ablator was bonded with a 1.12(5) μm glue layer (Hardmann Double-Bubble epoxy) to an 18(2) μm-thick square of Nb foil (Goodfellow 99.9% purity). Individual component and combined target thicknesses were determined by averaging over point measurements made using a Heidenhain length gauge with 0.1 μm precision, enabling determination of the glue layer thickness. For the A-targets, which comprised approximately half of the targets used, a ∼150 μm-thick single-crystal (100)-LiF window, with a 400 nm Al flash-coating, was attached to the rear surface of the Nb foil for pressure determination using impedance matching. The flash-coated side of the LiF crystals was bonded to the Nb using another ∼micron layer of Hardmann Double-Bubble epoxy. Figure 1(b) shows a schematic of the experimental setup at HED.

### B. Drive laser

Experiments at HED used the DiPOLE 100-X laser, an all-diode pumped 100 J-class ytterbium:YAG based laser with a frequency-doubled wavelength of 515 nm and a maximum repetition rate of 10 Hz [80,81]. However, during this experiment the laser was run at ∼0.02 Hz. The setup comprised a 250 μm phase plate, and the laser energy-polyimide pressure calibration was determined from polyimide-LiF shots preexperiment (Supplemental Material Sec. B [43]). Targets were driven using 10 ns quasi-flat-topped (125 ps rise time [87]) pulse shape, and a range of pulse energies from 4 to 49.5 J, to access single-shock states along the principal Hugoniot. Reproducible flat-top drive laser profiles and steady-state VISAR images after breakout revealed that the shock wave was temporally steady in the data presented here (shown in Supplemental Material Sec. K [43]).

At MEC, a Nd:glass optical laser, frequency doubled to a wavelength of 527 nm, was used with a 300 μm phase plate to focus a 10 ns quasi-flat-topped pulse shape into the ablator [88]. For the MEC experiments, we employed the same target A sample configuration as at HED and collected five shots with 80%–100% of the maximum available laser energy (100 J), thereby obtaining solid-liquid coexistence and liquid-only XRD data. At both facilities, the time delay between the laser drive and x-ray probe was tuned in each shot to ensure that the XRD data were obtained before shock breakout and release from the rear target surface.

### C. VISAR diagnostics

At HED, three VISAR channels were used to monitor the shock arrival time. An example image from one of the channels is shown in Fig. 1(c). The pressure was determined from VISAR-recorded particle velocities by matching VISAR fringe jumps for targets with a LiF window at the Nb-LiF interface of type-A targets or free-surface breakout at the rear of type-B targets (Supplemental Material Sec. C [43]). The VISAR field of view was ∼650 μm but a smaller subset region (∼100 μm) was used to determine particle velocities (Supplemental Material Sec. C [43]). These channels operated at 532 and 1064 nm with varying velocity-per-fringe configurations to determine unique particle velocities solutions giving accurate measurements between 10 m/s to 100 km/s [86]. At MEC, two streak cameras operating at 532 nm were used to collect VISAR measurements perpendicular to the Nb-LiF interface. The field of view in this instance was ∼250 μm.

### D. X-ray diffraction and density determination

At HED, a single pulse from the XFEL pulse train was delivered onto the sample with a focal spot size of 16 μm (V)×30 μm (H) (FWHM) at an angle of 22.5° to the target normal and VISAR axis [89]. The x-ray wavelength was 0.5166 Å and the sample to detector distance (SDD) was 317 mm. Two Varex large-area detectors [89], with a pixel size

of 150 μm × 150 μm, were employed to collect XRD images over a $Q$ range of 1.5 to 14.3 Å$^{-1}$ and azimuth range close to 180° [Fig. 1(b)], however, generally we were only able to resolve nonoverlapping shocked peaks up to 8.4 Å$^{-1}$. A $CeO_2$ standard placed at the sample position was used to refine the SDD and tilt of the detectors. The DIOPTAS [90] package was used to mask single crystal spots from LiF and integrate the XRD images into 1D profiles suitable for structure and density determination.

At MEC the x-ray beam was focused to 50 μm × 50 μm (FWHM) and centered on the focal spot of the drive laser which was aligned on the target. The x-ray wavelength was 0.7277 Å and the SDD was 687 mm. Similar to HED, Varex detectors were used at MEC spanning a $Q$ range of 2.2 to 11.1 Å$^{-1}$ allowing for high angle XRD coverage to be obtained up to the maximum angle. A $CeO_2$ standard was used for calibration, and XRD images were masked and integrated using the HEXRD [91] package.

All nonoverlapping diffraction peaks were fitted with Gaussians to determine their $2\theta$ position. The density of the compressed Nb was then calculated using four to seven solid diffraction peaks per profile, depending on the pressure, and a Nb atomic mass of 92.906 amu [92]. Fewer peaks were used to determine the density of the compressed state compared to the ambient pattern due to (i) peak overlap, (ii) reduced intensity of broadened peaks resulting from area conservation at high strain rates, and (iii) additional intensity loss at elevated temperatures as a consequence of the Debye-Waller factor, especially at large $Q$ [75]. Densities were corrected for variations in the SDD caused by manual mounting of targets and also the shot-to-shot x-ray beam jitter (see Supplemental Material Sec. F [43]).

## III. RESULTS AND DISCUSSION

### A. X-ray diffraction of niobium

An x-ray preshot was performed on each target prior to laser compression driven by flat-top drive laser profiles. In all reported data, the XFEL pulses probed the sample during shock transit (before shock breakout), thereby ensuring that XRD data were collected from only shock compressed volume and ambient (uncompressed) Nb in front of the shock. Figure 2 shows a waterfall plot of the 1D integrated diffraction profiles for selected runs collected at HED and MEC up to a $Q$ of 8 Å$^{-1}$. In preshot profiles, up to 13 diffraction peaks from bcc-Nb were visible, from which the ambient lattice parameter was determined to be $a = 3.3021(5)$ Å from HED and $a = 3.3040(5)$ Å from MEC, consistent with previous work [11].

Diffraction peaks from the uncompressed Nb ahead of the shock wave were observed in all runs presented here, as seen in Fig. 2. The diffraction peaks from the compressed material are observed to shift towards greater $Q$ with increasing pressure, as expected. Diffraction peaks from compressed Nb are significantly broader than those from uncompressed material due to large shock-induced strains [93,94] and variations in pressure states within large x-ray spot sizes [95]. In our experiment, these variations are expected to be smaller than the previously reported ∼2%−3% differences for a 50 μm × 50 μm XFEL x-ray spot with a 50 μm-thick polyimide ablator

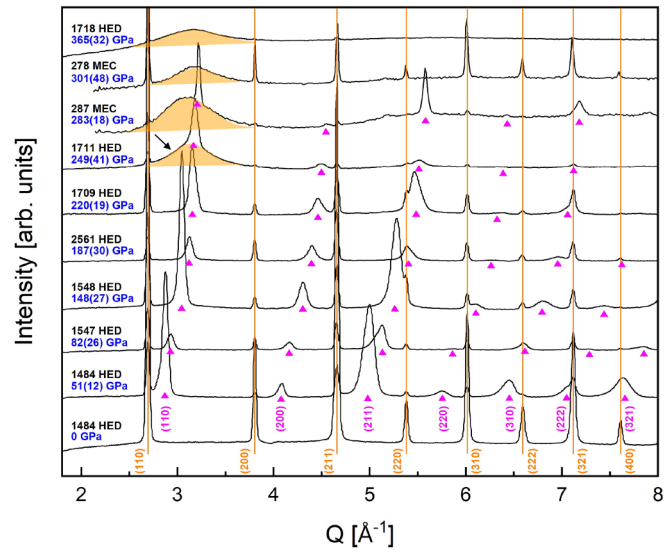



FIG. 2. Waterfall plot of the background subtracted integrated 1D XRD profiles from seven shots at HED and two at MEC. An ambient profile from Run 1484 HED is also shown. The pressure increases from bottom (0 GPa) to top (365 GPa). The black arrow highlights melting at P = 249(41) GPa (Run 1711 HED), as evident from the appearance of a diffuse scattering signal from the liquid phase. Shaded orange areas show a Gaussian peak, indicating the diffuse liquid scattering arising from the presence of melt. The sample was fully in the liquid state at 301(48) GPa. In each profile, ambient bcc-Nb peaks are identified by the orange lines while compressed-Nb peaks, if present, are identified by the pink triangles.

[95], due to employing a smaller x-ray probe at EuXFEL. We identify the onset of melting from the simultaneous weakening of bcc Bragg peaks and the emergence of liquidlike diffuse scattering around the (*110*) reflection, as can be seen at 249(41) GPa (Run 1711 HED) in the profile in Fig. 2. In other materials, such diffuse scattering has been reported as evidence of the liquid state [96–100]. Diffraction peaks from solid compressed Nb were not observed beyond 301 GPa (Run 278 at MEC—see Fig. 2), indicating that the sample is fully melted at this pressure. We obtained two shots in which the shocked Nb was fully molten, while weak peaks from uncompressed Nb remained from regions of the target ahead of the shock front confirming that the liquid signal arises from compressed material rather than a second stable solid phase at high pressure. The crystalline peaks in the liquid shots were confirmed to originate from ambient material, given the shock front transit time.

### B. Structure of niobium along the principal Hugoniot

Figure 2 shows that shocked Nb remains stable in the bcc crystal structure along the Hugoniot up to the liquid state. The background-subtracted 1D integrated profile of shock-compressed Nb at 220(19) GPa (Run 1709 HED), the highest-pressure datum obtained before the onset of melting, can be seen in Fig. 3.

A Le Bail fit [101] to the 220 GPa profile (Fig. 3) shows that both compressed and uncompressed material have the bcc structure, with lattice parameters of 2.814(11) and

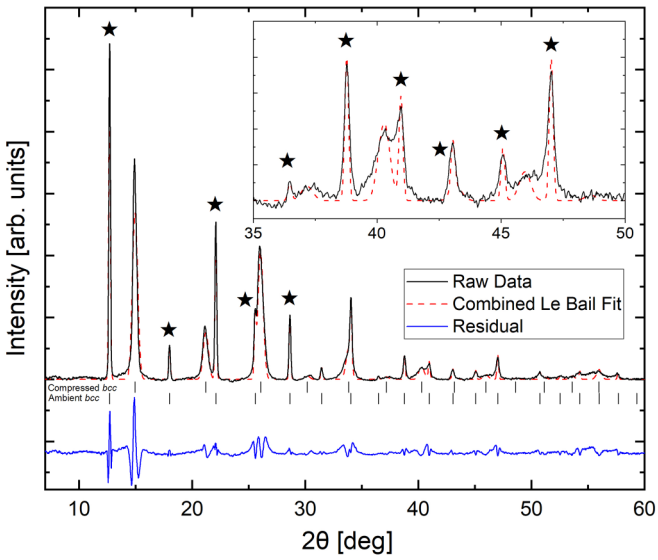



FIG. 3. The background-subtracted 1D diffraction profile from shocked-Nb at 220(19) GPa (Run 1709 HED). Stars have been added to highlight a number of low-angle peaks from uncompressed material. A two-phase Le Bail fit to the profile is shown as a red dashed line, confirming that the compressed material has the same bcc structure as that of Nb at ambient conditions. The residual to the fit is plotted as the blue line beneath the profile, and the insert provides an enlarged view of 35° to 50° range in $2\theta$.

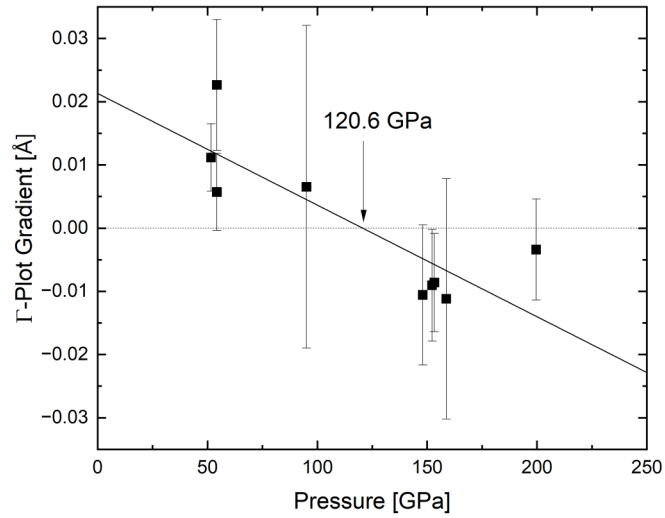



FIG. 4. The gradient of the Γ-plots as a function of pressure is shown for shots in which the $d$-spacings of the first five Bragg reflections could be measured. The gradient changes sign at 120.6 GPa, as a result of the change of sign of $S$, provided a positive $t$ value.

3.3017(4) Å, respectively. At higher pressures, in the solid-liquid coexistence profiles obtained along the melt curve, we also see diffraction from compressed bcc-Nb in addition to scattering from the liquid (Fig. 2). We observe no evidence of the previously predicted bcc → hcp (∼75 GPa) or bcc → *Pnma* (∼142 GPa) phase changes in Nb along the principal Hugoniot [23,42,68]. Simulated profiles from the hcp and *Pnma* structures at 95 and 148 GPa, respectively, show that neither can explain the observed diffraction profiles at these pressures (Fig. S5). Furthermore, none of the *additional* diffraction peaks predicted by these structures is observed, enabling us to rule out the presence of both, even as a minority phase, above a volume fraction of 1.5%. The bcc crystal structure of Nb presented here along the principal Hugoniot is consistent with previous static compression studies up to 145 GPa [6,11–14].

### C. Strength analysis

We found that the standard deviations in lattice parameters determined from the positions of individual XRD peaks within a given profile typically differed by 0.005 Å, and in one instance by up to 0.012 Å. This suggests the presence of residual elastic shear strain (i.e., strength) [93]. Variations of this magnitude are consistent with those expected in laser-driven shock compression experiments [93,94,102,103]. However, attempts to quantify the deviatoric stresses using the analysis of Singh and co-workers [104–106] were unsuccessful due to limited detector coverage and significant texture making it difficult to extract strain vs azimuth data around the Debye-Scherrer (DS) rings. Hence, we employed Singh's analysis using Γ-plots to determine the macroscopic deviatoric stress ($t$) along the principal Hugoniot (see Supplemental Material Sec. G [43]). Singh's method becomes unstable near 130 GPa for Nb, where small and rapidly changing values of the elastic anisotropy, $S = S_{11} - S_{12} - (S_{44}/2)$, (see Fig. S9) can lead to large, unphysical values of $t$. As a result, we were unable to extract meaningful information to quantify the macroscopic deviatoric stress. However, Fig. 4 shows the gradient in the linear fits of the Γ-plots changes sign at 120.6 GPa, as expected for a change in sign of $S$ (assuming $t$ remains positive). This estimated crossing pressure is in good agreement with the 130 GPa predicted from the elastic constants calculations of Wang *et al.* [107]. Our Singh analysis therefore provides experimental evidence for a sign change in $S$ at 120.6 GPa, and consequently for the transition from heat-induced hardening to heat-induced softening described by Wang *et al.* [107], provided $dC_{44}/dP \gtrsim 0.28$. Wang *et al.* report a value of $dC_{44}/dP = 0.88$ along the principal Hugoniot.

### D. Comparison to previous studies

The resulting pressure-density curve for shock-compressed Nb is shown in Fig. 5. The pressure-density Hugoniot behavior we observe in this work is consistent with previous shock compression studies on Nb in the solid phase up to 249 GPa along the principal Hugoniot [25,26,67,108–110]. Using static compression and laser-heating techniques, Errandonea *et al.* [23] reported a phase transition from cubic-bcc to orthorhombic-*Pnma* and predicted this transition to also occur along the principal Hugoniot. We do not observe any departure from the ambient-pressure bcc structure under our experimental conditions in the solid and solid-liquid coexistence regions along the principal Hugoniot. Kinetic effects due to the compression rate have previously been shown to influence structural transformations and can shift, or completely eliminate, phase transition boundaries in metals under dynamic vs static compression [97,111–113]. However, another possible solution is that under extended heating, Errandonea

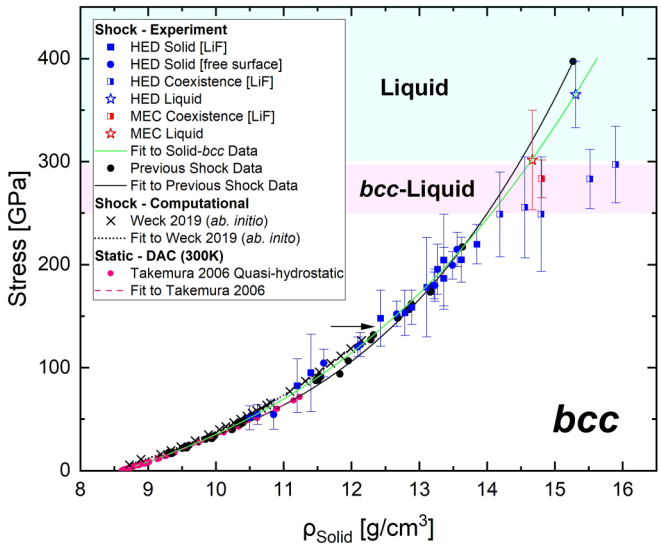



FIG. 5. The compression curve of bcc-Nb along the principal Hugoniot from data collected at HED (blue symbols) and MEC (red symbols). The green curve shows the fit to all our solid data, extrapolated to 365 GPa. The half filled squares indicate bcc-liquid coexistence and the two unfilled stars indicate fully molten Nb. For the two liquid shots the density was estimated from the extrapolated solid Hugoniot. The black crosses show the data obtained from the molecular dynamics simulations of Weck *et al.* [118], while the filled black circles show previous shock compression data [25,26,42,67,108–110]. The quasihydrostatic static compression data from Takemura *et al.* [11] at 300 K are shown as pink circles. The horizontal black arrow at 142 GPa identifies the predicted solid-solid phase transition pressure to *Pnma*-Nb from Errandonea *et al.* [23]. The pink shading shows the minimum bound for the bcc-liquid coexistence region, which starts at 249 GPa and ends at 298 GPa, while the blue shading shows the liquid region above 301 GPa, the pressure at which the fully melted Nb was first observed at MEC. The average error on the pressure and density is 25 GPa and 0.02 g/cm$^3$ respectively for our data from HED and MEC.

*et al.* formed a Nb-carbide due to carbon diffusion from the diamond anvils. Their published results show no evidence that a single phase, *Pnma*-only, diffraction profile was obtained, perhaps suggesting partial reaction of the sample. Although *Pnma* symmetry has been reported in $Nb_2C$ [114–116], its lattice parameters are different to those reported by Errandonea *et al.* However, the Z-method used by Errandonea *et al.* has previously been shown to struggle to resolve temperature induced solid-solid transitions [117], while the recent *ab initio* calculations by Wang *et al.* [107] have shown the proposed high-$T$ *Pnma* structure to be mechanically unstable under hydrostatic conditions. Repeating the calculations of Errandonea *et al.* resulted in significant stress anisotropy, suggesting that the *Pnma* structure may only achieve metastability under non-hydrostatic conditions [107].

Al'tshuler *et al.* [42] interpreted a change in the $U_s$-$U_p$ slope at 94 GPa as a second-order phase transition. Scatter in our $U_s$-$U_p$ data makes it unclear whether such a slope change is observed in this study (Supplemental Material Sec. D [43]). We find limited evidence for this transition, with no anomalies observed along the compression curve in Fig. 5.

Incipient melting of Nb was observed to start at 249(41) GPa (Run 1711 HED) and was complete at 301(48) GPa (Run 278 MEC), Fig. 2. Recent *ab initio* calculations by Wang *et al.* [107] reported melting of Nb to begin at 207 GPa (6850 K) and reach the liquid state at 255 GPa (7400 K), around 50 GPa lower than the experimental values presented here. Therefore, our data suggest a steeper melt curve than previous studies [23,107,119]. We observed profiles of molten Nb at 301(48) GPa at MEC and 365(32) GPa at HED. As no density for liquid-Nb was determined, the liquid data points (unfilled stars in Fig. 5) have been plotted using densities estimated from the extrapolated solid Hugoniot and pressures determined from VISAR.

Within the solid-liquid coexistence region, we obtained five shots at HED, and one at MEC. Interestingly, around the melt transition the VISAR velocities increase weakly with increasing drive laser power, while the density of the solid component continues to increase significantly. This leads to an apparent softening away from the extrapolated solid Hugoniot (green curve) in Fig. 5.

We attribute the deviation observed in the pink-shaded coexistence region of Fig. 5 to the following. For phase boundaries with a positive Clapeyron slope [120], such as in Fe [99], the solid phase is denser than the coexisting liquid at a given $(P, T)$. Consequently, densities determined from the solid component are expected to lie at higher density than the bulk equation of state (represented by the linear $U_s$-$U_p$ model, which should not differ to a large extent in the incipient melting regime from that established in the solid regime). Within the coexistence region, the Hugoniot density corresponds to a volume-fraction-weighted average of the coexisting solid and liquid densities. The solid densities shown in Fig. 5 (half filled squares) therefore lie on the solidus, rather than on the Hugoniot, hence their increased density relative to the extrapolated solid Hugoniot (green curve), while remaining on the melt curve.

## IV. CONCLUSION

In conclusion, through a combination of experiments using the DiPOLE 100-X laser at the HED instrument of the EuXFEL and the long-pulse laser at the MEC end station of the LCLS, we have shock compressed Nb along the principal Hugoniot and observed the persistence of the low-pressure bcc structure up to 220 GPa before the onset of melting. By combining multiple shots carried out at both facilities, we present XRD data from 51 GPa (bcc) to 365 GPa (liquid). We find that the principal Hugoniot starts to intersect the Nb melt curve at 249 GPa, and the transition to the pure liquid phase is complete at 301 GPa, suggesting a steeper melt curve than previously predicted. Based on our *in situ* XRD measurements, we do not observe any previously reported phase transitions and Nb remains stable in the bcc structure up to liquid.

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## DATA AVAILABILITY

The data is available upon reasonable request at Ref. [121].


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